Turgor Pressure Moves Polysaccharides into Growing Cell Walls of Chara corallina

TIMOTHY E. PROSEUS and JOHN S. BOYER*
College of Marine Studies and College of Agriculture and Natural Resources, University of Delaware, 700 Pilottown Road, Lewes, DE 19958, USA

Received: 12 August 2004 Returned for revision: 4 October 2004 Accepted: 26 January 2005 Published electronically: 10 March 2005

INTRODUCTION

The following work was undertaken to investigate whether turgor pressure (P) affects the deposition of polysaccharides in plant cell walls. Young plant cells capable of growing must build more wall as they grow. Hemicellulose, pectins and small amounts of structural protein form the wall matrix in which cellulose microfibrils are usually embedded to create a reinforced gel structure (e.g. Morrison et al., 1993). The wall grows in part by being stretched by P because interpolymeric bonds in the matrix naturally break and re-form, and when under tension from P the polymers tend to slip past each other irreversibly, enlarging the wall (Cleland, 1971; Taiz, 1984). Typically, the stretching enlarges the cell 10- to 100-fold (Roberts, 1994) but under normal conditions the walls do not become proportionately thinner (Bonner, 1934; Loescher and Nevins, 1973; Bret-Harte et al., 1991). The wall increases in mass and volume while continuing to support the protoplast and resist the strain produced by P. As much as 90–99 % of the wall is new by the time the cell matures (Roberts, 1994).

Initially, the new wall materials accumulate at the inner wall face after being secreted through the plasma membrane by exocytosis (matrix component of wall) (Northcote and Pickett-Heaps, 1966; Ray, 1967; Moore et al., 1991; Perrin et al., 2001) or by plasma membrane complexes (cellulose microfibril component of wall) (Giddings et al., 1980; Kimura et al., 1999). Once released, the new matrix polymers find their way into the growing wall and around the new cellulose microfibrils (Northcote and Pickett-Heaps, 1966; Ray, 1967; Preston, 1974; Carpita and Gibeaut, 1993; McCann and Roberts, 1994). Within the matrix structure, a considerable fraction of the volume is water-filled and forms a porous network for molecule movement (Thompson and Preston, 1968; Carpita and Gibeaut, 1993; McCann et al., 1995; Kohorn, 2000). The network allows molecules as large as 3.5–8.6 nm to enter at atmospheric P (references in Read and Bacic, 1996; Titel et al., 1997).

In some respects, the matrix might act as a gel filtration system similar to cross-linked dextran such as Sephadex that separates polymers by molecular size (Laurent and Killander, 1964). Tepfer and Taylor (1981) found this type of behaviour with cell wall particles placed in a column through which a polymer solution flowed. However, living cells develop P on one side of the wall. A tension is created that causes the coiling, intercalating network of matrix molecules to partially uncoil and straighten, restricting polymer motion and causing a more uniform conformation than in the unstrained matrix. In primary walls capable of growth, this conformational change is seen as an elastic enlargement of the matrix followed by irreversible deformation as P becomes higher (Taiz et al., 1981; Proseus et al., 1999). The P also creates an abrupt drop in pressure at the inner face of the wall.

Key words: Dextran, FITC–dextran, fluorescein isothiocyanate, gold colloids, turgor pressure, wall matrix.
wall face that forms a steep gradient \( dp/dx \) extending into the matrix only a short, sub-polymer distance \( x \). The \( P \) step-down represents energy that could affect nearby molecules.

The effect of gel filtration, wall strain and steep \( P \) gradients could be important for new wall formation in growing plant cells. In the present work, this possibility was investigated in walls isolated from large algæ cells. \( P \) was produced with a method adapted from the mercury injection procedures of Kamiya et al. (1962, 1963) and Metraux et al. (1980) but using mineral oil injected into the lumen created by removing the protoplast. Various polymers or colloids were pushed ahead of the oil to the inner face of the wall as though they were recently secreted by the plasma membrane. Their movement through the wall was then monitored at various \( P \).

### MATERIALS AND METHODS

#### Plant material

Cultures of *Chara corallina* (Klien ex. Willd., em. R.D.W.) were maintained as previously described (Zhu and Boyer, 1992). Briefly, *C. corallina* was grown in a liquid medium with a sand and pond mud substrate. The medium was maintained at 23 °C and in continuous light (PAR of 10 \( \mu \)mol m\(^{-2}\) s\(^{-1}\) at the medium surface). Macronutrients (1 mM chloride salts of Na, K, Ca and Mg) were provided with periodic medium changes. The pH naturally settled at 5.4 during all cell manipulations unless otherwise noted.

#### Isolating cell walls

For each experiment the ultimate or penultimate internode cell was excised by hand from the distal portion of the thallus by cutting below the cell’s basal node and trimming away any nodal branch cells. These internode cells had been growing actively, ensuring that only primary cell walls were present. After excision, a stream of warm air (cell at 30 °C) was used to reduce the turgor, and the basal wall end was cut off leaving a cell about 15 mm long. After transferring to a shallow dish of water, the intact end of the cell was held stationary under the water with wax-coated forceps, and the remainder of the cell was gently scraped with a blunt plastic blade until no green colour remained. Under an electron microscope, the chloroplasts could be seen closely associated with the plasma membrane in a thick, parietal layer of cytoplasm. Removing the green colour removed this layer with the chloroplasts and plasma membrane.

#### Preparing and pre-loading microcapillaries

A borosilicate capillary (1 mm OD) was hand-drawn in a flame to produce a tapered hollow microcapillary tip that was trimmed to approx. 200 \( \mu \)m diameter and fire-polished to remove sharp edges. The capillary was filled with mineral oil (paraffin oil) and fitted into a large cell pressure probe (Steudle and Zimmermann, 1974) also filled with mineral oil. The probe used by Proseus et al. (2000) was modified to include a precision motor (Oriel Co., Stratford, CT, USA) to drive the steel plunger into the probe barrel using a programmable electronic feedback system (model DP25-EAR, Omega Engineering, Stamford, CT, USA). The system allowed the control of \( P \) to within 0.001 MPa measured with a Kulite pressure transducer (Model CQ-140-200-D, Kulite Semiconductor Products, Inc., Ridgefield, NJ, USA). After pre-loading with oil and mounting on the pressure probe, 3 \( \mu \)L of one of several test solutions (see Injecting polymers below) could be loaded into the microcapillary by immersing the tip in a droplet of the solution, and pulling the droplet into the tip. Because of the large diameter of the microcapillary tip, the droplet entered easily without surface tension effects. Care was taken to avoid air bubbles in the tip or pressure probe.

#### Sealing isolated cell walls to microcapillaries

The micromanipulator holding the pressure probe was adjusted to immerse the distal 5 mm of the microcapillary tip into a shallow dish of water. The open end of the isolated cell wall was threaded over the tip until the wall and glass formed a tight junction at least 2 mm long around the entire circumference of the tip. The probe was repositioned to bring the glass/wall junction just above the water surface while leaving the free end of the wall lumen (approx. 10 mm) submerged. The glass/wall junction was allowed to air-dry for a few minutes, then acetone was applied and allowed to dry again. The junction was sealed with cyanoacrylate glue (‘Quick Tite’, Locitite, Rockyhill, CT, USA) by applying a drop of glue and blotting the excess, thereby allowing the glue to penetrate on all sides, creating a clear, colourless junction. The glue was applied three times, curing 15 min after the first and second times and 1 h after the third. Water in the junction was indicated by a dull white appearance and destroyed the sealing abilities of the glue, a condition that could not be reversed by applying more glue. When the cure was complete, the probe was raised to lift the entire cell wall above the water level. The cell wall was dipped in clean water to remove glue residue and lowered into a collecting vessel whose size and contents varied depending on the particular experiment.

#### Injecting polymers

Molecules released by the plasma membrane (PM, Fig. 1A) are in the periplasmic space (yellow circles, Fig. 1A). In an isolated wall mounted on the microcapillary tip, the pre-loaded test solution could be pushed into the same ‘periplasmic space’ by the mineral oil in the pressure probe (yellow circles, Fig. 1B). The nonpolar mineral oil was unable to pass through the cell wall because surface tension at the oil/water interface prevented the oil from entering the aqueous phase of the wall matrix (Fig. 1B). The same force controls the position of the plasma membrane in the living cell (Fig. 1A). The inability of the oil to pass through the wall caused pressure to build in the injected oil and created an ‘artificial’ \( P \) (Fig. 1B). The injection rate was controlled to keep the \( P \) from rising too abruptly (usually requiring 2–5 min) after which \( P \) was adjusted to a
final value between 0.05 and 0.5 MPa, and the programmable control system was set to maintain P constant. By this method, the test solution could be subjected to P identical to that in the living cell.

Aqueous test solutions were prepared containing unconjugated fluorescein isothiocyanate (FITC) (0.1 mg mL⁻¹), or one of four preparations of conjugated FITC–dextran (α-1,6-glucan with occasional α-1,3- and α-1,4-links, polymer about 5 % α-1,3-branched, 6-0 mg mL⁻¹) having an average molecular weight of 4-4 kDa, 38 kDa, 70 kDa or 580 kDa (lot numbers FD-4 (35H1368), FD-40 (78H0457) and FD-70 (128H9801), FD-500 (74H5012), respectively, from Sigma, St Louis, MO, USA). Test solution stocks were stored at 4 °C and discarded after 7 d.

**Measuring polymer movement into and through cell walls**

For most trials, measurements began when all of the 3 μL of test solution was inside the lumen previously occupied by the protoplast, and a stable target P had been reached (time = 0). After stirring the external water bathing the wall, an aliquot was periodically removed and its fluorescence intensity measured in a Shimadzu RF-1501 spectrofluorophotometer (Shimadzu, Kyoto, Japan) with excitation at 484 nm and emission at 515 nm. Trials were allowed to continue until the intensity of the external water stopped increasing, indicating no more test solute was passing through the wall. Data were expressed as percentage recovery of the solution supplied to the lumen, based on the fluorescence intensity of a like quantity of test solution but measured separately. The measurements indicated there was a 3-8 % loss of test solution during tip loading, indicating no more test solute was passing through the wall. The measurements indicated there was a 3-8 % loss of test solution during tip loading.

The experiment room was kept dark except during brief periods when sample aliquots were being taken or glue was being applied. Three replicate trials with each test solution were performed at each of up to four target P.

In another set of experiments, the location of FITC or FITC–dextran in the lumen, wall matrix or external water was determined by injecting FITC or 38-kDa FITC–dextran in the usual way and developing a typical P of 0.5 MPa. After P developed for 1.5 min, the fluorescence in the external water was measured to indicate the amount that had exuded through the wall. The wall was then cut into several fragments with a razor blade, and test solution (and mineral oil) within the cell lumen was flushed into the external water during the cutting. The external water was thoroughly stirred with a pipette for 15 s and the additional fluorescent intensity was measured. This sample indicated the amount of test solution originally exuded through the wall plus that collected from the lumen. Then, the wall fragments in the water containing the exuded + lumen contents were immediately placed into a Ten Broeck tissue grinder and ground for several minutes until no visible fragments remained. This extract was centrifuged for 10 min at 700 g, and the additional fluorescence was measured in the external water to indicate the amount in the wall matrix. The sum of fluorescence in the external water (i.e. exuded + lumen + wall matrix) was compared with that originally injected into the cell lumen. Care was taken to keep the samples in darkness as much as possible prior to fluorescence measurements. Early trials showed that mineral oil escaping into the bathing water during cutting did not disturb the fluorescence measurements.

**Microscopy**

Observations, using a Zeiss LSM510 laser scanning confocal microscope (Carl Zeiss, Thornwood, NY, USA), were made of cell walls with artificial P using FITC-based test...
solutions in unstirred, single-chambered walled cover-glasses (Nalge Nunc, Naperville, IL, USA). The isolated walls were stained with 0.1% aqueous Congo Red solution (Sigma Chemical Co., St Louis, MO, USA) for 30 min prior to mounting on the microcapillary of the pressure probe. Dual channel time-series images of the injection, pressurization, and movement of fluorescent FITC, 38-kDa FITC–dextran and 580-kDa FITC–dextran were made with excitation at 488 nm and emission at 505–530 nm controlled with a bandpass filter. Wall-bound Congo Red was simultaneously imaged with excitation at 543 nm and emission at 560 nm with a longpass filter. In the final images, FITC signal was false-coloured yellow.

Supplying gold colloids

In another group of experiments, 4 μL of aqueous suspensions of colloidal gold with nominal diameters of 5 nm, 10 nm or 20 nm (ICN, Aurora, OH, USA) were injected into the cell lumen. The procedures were the same as with FITC–dextran solutions except the external water was replaced with microscope objective immersion oil. Walls were blotted with absorbent paper before being lowered into the oil. Pressurizing the wall lumen forced the aqueous test suspension through the wall and formed droplets on the wall exterior or the chamber bottom. The droplets were collected with a Drummond microdispenser, transferred to water in a disposable cuvette, and stirred. Light transmission was measured at 514 nm (5-nm colloids) or 524 nm (10- and 20-nm colloids) with a Hewlett-Packard 8452A UV/VIS spectrophotometer. This method allowed for only one reading per wall, so trials were repeated and the duration at the target \( P \) was varied to establish a time course. Standard curves established for each stock suspension showed that light transmittance at the characteristic wavelength was inversely proportional to the concentration of gold colloid. Colloid diameters were verified by transmission electron microscopy (TEM). Formvar-coated 400-mesh grids were floated on a drop of colloidal gold (on Parafilm) for 10 min and then air-dried. The TEM images indicated that the colloidal particles were highly uniform with the nominal diameter of 5-nm particles measuring 4.6 ± 0.1 nm (range) and the 10 nm particles measuring 9.8 ± 0.1 nm (range). All colloidal suspensions were stored at 4 °C at pH 7.5–8.0 as supplied by the manufacturer, and the stock solutions were discarded after 3 weeks. Control walls were treated as above, except 4 μL of water containing no gold colloid was injected.

RESULTS

Injecting test solutions into the lumen

When the pressure probe injected mineral oil into the lumen of the isolated wall, artificial \( P \) could be generated and precisely controlled. Placing an aqueous test solution of FITC–dextran in front of the oil allowed the trans-wall movement of the solute to be monitored at various \( P \). In the confocal microscope, the entering FITC–dextran solution was dilute (lumen, Fig. 2A) and became concentrated (brighter fluorescence) as \( P \) began to develop (Fig. 2B). When \( P \) approached 0.5 MPa, the FITC–dextran was in a very small volume, highly concentrated in the lumen or wall matrix. Only a slight amount of FITC–dextran was visible in the external water next to the wall (Fig. 2C). After a few minutes, FITC–dextran began to be detected outside (Fig. 2D).

This process was quantified by detailed measurements of water movement, fluorescence and \( P \) during the pressurization. Water was collected from the external surface of the wall when surrounded by a saturated atmosphere while \( P \) was developing. As oil began to fill the isolated wall, \( P \) rose but then decreased as oil injection was paused briefly (Fig. 2E, steps in \( P \) before 0 min). The water in the test solution flowed out of the lumen and across the wall where it accumulated as droplets on the outside of the cell (Fig. 2E, W before 0 min). Oil had to be repeatedly injected to compensate for this water movement. When the 38-kDa FITC–dextran was measured in the water outside of the cell, little could be detected (Fig. 2E, F-d before 0 min). The FITC–dextran remained mostly in the lumen or wall, confirming that dextran had been concentrated while water was moving out. At high \( P \), nearly all of the water in the dextran solution had moved through the wall (98% of solution supplied to interior was collected outside as water; Fig. 2E, W at 0 min). Water flow ceased, and bulk flow through the wall ended. Only after bulk flow ended was it possible to bring \( P \) to its final level where it was stable (Fig. 2E, stable \( P \) after 0 min). At this high \( P \), dextran slowly began to appear externally (Fig. 2E, F-d after 0 min). The fluorescent signal became stronger with time as more FITC–dextran moved through the wall and into the external water.

This movement of FITC–dextran was affected by \( P \). Initially, the FITC–dextran was forced against the inner face of the wall and, after 5 min at \( P \) of 0.5 MPa, was visible inside the lumen and outside after having passed through the wall (Fig. 3A). In a separate cell, 25 min at a lower \( P \) of 0.05 MPa failed to move the FITC–dextran through the wall (Fig. 3B).

In order to test whether the rate of dextran movement was affected by the size of the molecule, different molecular weights of dextran were injected into isolated walls, and \( P \) was kept the same for each experiment. The initial injection of dextran and its subsequent movement through the wall to the external water was observed continuously in the confocal microscope. The appearance of fluorescent signal in the external water at \( P \) of 0.5 MPa was more rapid for FITC (Fig. 4A) than for FITC–dextran having a molecular weight of 38 kDa (Fig. 4B). If the FITC–dextran had a molecular weight of 580 kDa, it scarcely moved through the wall (Fig. 4C).

Polysaccharide movement through cell walls

Because both \( P \) and molecular size affected polymer movement through the walls, the effects were quantified more fully by monitoring the trans-wall movement over a wide range of \( P \) and molecular sizes. When FITC alone was supplied to the lumen, only about 10–40% of it was detected in the external water after 0.5 min of stable \( P \) (Fig. 5A). It then passed slowly to the outside over the next hour. There was no difference in FITC movement at \( P \) of 0.05 MPa and 0.5 MPa, and the slightly slower movement...
at 0.25 MPa was not statistically significant. Between 80 and 90% of the FITC injected into the cell lumen was detected outside of the wall after 120 min.

When FITC was conjugated with 4-4 kDa dextran, it moved more slowly through the wall than the FITC, and only 0–10% of the dextran was detected outside after 1 min of stable \( P \) (Fig. 5B). Subsequently, the dextran moved through the wall until 70–93% was found outside at 3–5 h. The movement was fastest at 0.5 MPa and slowest at 0.05 MPa.

With 38-kDa FITC–dextran, there was little trans-wall movement near time zero (Fig. 5C) or during the entire experiment at \( P \) of 0.05 MPa. At 0.25 MPa, the 38-kDa FITC–dextran crossed the wall slowly. At 0.5 MPa, movement across the wall increased greatly and 80% was recovered after 300 min.

With 70-kDa FITC–dextran, trans-wall movement was negligible at \( P \) of 0.05 MPa and 0.25 MPa (Fig. 5D). At 0.35 MPa, movement occurred slowly and at 0.5 MPa.
much faster. After 300 min, 74% of the latter dextran was recovered. The lack of movement at $P$ of 0/5 MPa and 0/25 MPa indicated that a threshold $P$ was necessary before the trans-wall movement began.

With 580-kDa dextran, movement was undetectable at $P$ as high as 0/5 MPa (Fig. 5E) even after 250 min. Controls in which only water was injected as a test solution produced no fluorescent signal in the external water (not shown).

Dimensions of interstices in cell walls

Because $P$ could move some of the dextrans through the wall, it was of interest to determine the dimensions of the path. The lumen was supplied with gold colloids instead of dextran, and their movement was determined in the same way as with the dextrans, but using a spectrophotometer. Colloids with a diameter of 4.6 nm were able to pass through the walls at $P$ of 0.5 MPa (recovery of 70% after 45 min), but their movement was significantly lower at $P$ of 0.25 MPa (recovery of 10% after 60 min) (Fig. 6A). As one stock vial of nominal 5-nm colloid aged it suddenly became much less able to penetrate the cell walls at any $P$. Subsequent TEM images showed that the colloid had coalesced into trimeric and tetrameric clusters that were too big to pass through the walls. Colloids with diameters of 20 nm and 9.8 nm scarcely moved and were detected outside the wall at about 10% recovery after 60 min at $P$ of 0.5 MPa (Fig. 6B and C). Attempts to explore the movement of 2.6-nm Nanogold spheres (Nanoprobes, Inc, Yaphank, NY, USA) were unsuccessful because the particles bound strongly to the wall allowing only a small fraction of them to move through at any $P$ (data not shown).

The $P$-responsiveness with gold of 4.6 nm in diameter and some of the dextrans raised the possibility that high $P$ irreversibly enlarged the interstices in the wall matrix. To test this possibility, $P$ was kept at 0.5 MPa for a few minutes and returned to 0.05 MPa while 38-kDa FITC–dextran was moving through the walls. The sudden lowering of $P$ (Fig. 7A, Wall 2) rapidly halted the movement of the FITC–dextran (Fig. 7B, Wall 2), as compared with the typical curve with continuous high $P$ (Fig. 7, Wall 1). Three replicate trials of the experiment in Fig. 7 gave similar results.

Location of polymers

It was of interest to determine whether the concentrated solution of dextran was inside the lumen or inside the wall matrix when stable $P$ was first achieved. The fluorescence of FITC or FITC-conjugated dextran was measured in a) a sample of the external bathing water (Fig. 8, Exuded), (b) the solution flushed out of the lumen after excising the end of...
the wall (Fig. 8, Lumen) and (c) the remaining wall, which was extracted in water (Fig. 8, Wall). With FITC alone, 62% had exuded into the external bathing water when the test solution had been pressurized at 0.5 MPa for 1.5 min (Fig. 8A). The lumen contained 8%, and the wall contained 30% (Fig. 8A). This result indicated that FITC had been rapidly flushed into the wall matrix from which it diffused into the external water.

In contrast, large molecules such as 38-kDa FITC–dextran did not enter the wall matrix rapidly, and cutting the wall open recovered nearly 80% of the dextran after 1.5 min of pressurization (Fig. 8B, Lumen). This indicated that only a minor fraction of the dextran was within the wall matrix at 1.5 min. Even after 15 min at P = 0.5 MPa, 40% of the dextran was still trapped inside the lumen.

**DISCUSSION**

**Ultrafiltration and P effects on polymer movement**

The fate of polymers under pressure could be followed as they encountered the primary wall. The polymers at first were concentrated inside the wall, which acted
The swelling pressures were negligible). In multicellular plants down to atmospheric pressure in the wall (assuming inner wall face. As \( P \) protoplast concentrates the new soluble polymers at the inner wall face together with the steep gradient across the plasma membrane itself. The secreted polymers move through the abrupt step-down at the \( P \) layer of secreted polymers, and there is no difference in the matrix. In \( P \) to be flushed through the wall. Instead, it was driven diffusionaly by the steep concentration gradient across the inner wall face together with the steep \( P \) gradient across the inner wall face.

After the concentrating phase, the dextran moved into and through the wall at rates determined by polymer size and the magnitude of \( P \). The movement occurred despite the absence of bulk flow through the wall. Without bulk flow, there was no solute drag that would cause the polymer to be flushed through the wall. Instead, it was driven diffusionaly by the steep concentration gradient across the inner wall face together with the steep \( P \) gradient across the inner wall face.

These results indicate that the plasma membrane in living cells is displaced from the inner wall face by the intervening layer of secreted polymers, and there is no \( P \) difference across the plasma membrane itself. The secreted polymers are in a region of uniformly high \( P \), but as they enter the wall matrix they move through the abrupt \( P \) step-down at the wall face to the low \( P \) in the matrix. In \( Chara \), the \( P \) stepped down to atmospheric pressure in the wall (assuming swelling pressures were negligible). In multicellular plants whose cell walls contain water below atmospheric pressure, the \( P \) step-down would tend to be larger than in \( Chara \).

**Dimensions of matrix interstices**

All of the dextrans displayed long residence times in the lumen and matrix. The dextran contained no charged groups, and those passing through the wall eventually were 80–95 \% recovered in the external water indicating that they moved slowly but without actually becoming part of the wall matrix.

The pre-existing matrix contained water-filled interstices having dimensions reported to be controlled mainly by the pectin fraction (Baron-Epel et al., 1988; Meiners et al., 1991; McCann et al., 1992; Fleischer et al., 1999) that microscopy showed to have a regular network of linkages (Casero and Knox, 1995). Consequently, the diffusion path should consist of tortuous interstices with a narrow range of dimensions, at least within comparable wall regions (McCann et al., 1990). As long as \( P \) was low in the present work, the wall appeared to act by typical gel filtration (Laurent and Killander, 1964; Tepfer and Taylor, 1981). Small molecules had greater wall access than large molecules. However, when \( P \) was increased, this model failed. The rate at which small molecules moved through the matrix was unaffected, but large molecules moved faster than at low \( P \).

There was no evidence of specialized pores in the wall matrix other than plasmodesmata that existed only at the internode ends in contact with the small nodal cells (Meeuse, 1957). The nodal cells remained generally intact but the plasmodesmatal connections appeared inoperative because polymer movement through the ends was negligible compared with that through the side walls. In general, molecular movement through the side walls was uniform.

Gold colloids were more reliable than dextran for measuring the dimensions of the matrix interstices because the solid gold spheres were dimensionally stable and were verified by TEM. For a given value of \( P \), the wall either had openings large enough to admit a given colloid or it did not. The wall admitted colloids of 4-6 nm but only at a high \( P \) of 0-5 MPa. At slightly lower \( P \), the gold scarcely entered, suggesting that the concentration of gold colloid was less than at high \( P \) or the interstices had become smaller, or both. The highest \( P \) used in the experiments (0-5 MPa) was within the range normally causing irreversible cell enlargement in living \( Chara \) internodes (Proseus et al., 2000). It seemed possible that the matrix interstices became similarly enlarged at high \( P \) but not lower \( P \). However, using dextrans, the rapid polymer movement at high \( P \) immediately returned to the low-\( P \) rate when \( P \) decreased. Had high \( P \) irreversibly enlarged the openings in the matrix, the rate would have remained rapid. The rate of gold colloid movement was thus affected either by changes in dimensions that were reversible or by concentrating effects at the inner wall face.

The FITC results argue against reversible changes in dimensions. FITC was small enough (0-8 nm) to be flushed into the matrix while water moved through prior to time 0. As a result, the FITC was outside of the \( P \) boundary, and \( P \) did not act to concentrate the molecule or deform it. \( P \) acted on FITC movement only by straining the matrix. The constancy of FITC diffusion in the matrix at all \( P \) indicates that straining the matrix did not enlarge wall interstices enough.
to affect the diffusion of small molecules. These principles probably govern the movement of all small molecules into and through the wall matrix. The FITC results, together with the gold colloid data, suggest that the molecular exclusion diameter was not much above 4\(\times\)10\(^{-6}\) nm.

These dimensions were similar to those reported in *C. corallina* by Shepherd and Goodwin (1989) who estimated the maximum molecular diameter to penetrate the walls at about 4\(\times\)10\(^{-6}\) nm in turgorless cells using a rhodamine-labeled, globular oligopeptide, insulin A. Berestovsky et al. (2001) reported sizes of only 2\(\times\)10\(^{-6}\) m in *C. corallina* based on P changes caused by PEG (polyethylene glycol) movement across the walls, but these authors did not adequately account for the low reflection coefficient of the walls for PEG. The low reflection coefficient would prevent P from fully accounting for the osmotic effect of the entering PEG, thus leading to an underestimate of molecular exclusion.

Given that diameters in the matrix interstices were approx. 4-6 nm in the *Chara* walls, FITC with its molecular weight of 389 and diameter of 0-8 nm (Ekani-Nkodo and Fygenson, 2003) moved readily through the wall, while the 4-4-kDa FITC–dextran had a Stokes’ diameter of 3-5 nm (Peters, 1986; Read and Bacic, 1996) and began to be hindered. All the other dextrans had Stokes’ diameters greater than 4-6 nm and did not enter at low P, i.e. 38-kDa dextran (diameter 9 nm), 70-kDa dextran (11 nm) and 580-kDa dextran (27 nm) were strongly hindered or did not move through the wall detectably. As a consequence, gold colloid at high P and FITC–dextran at low P gave similar dimensions for matrix interstices (Fig. 9A and B).

**Movement of large molecules into small openings**

At high P, however, 38-kDa and 70-kDa dextran moved through the wall. The larger of these had a Stokes’ diameter of 11 nm or about twice the gold-derived diameter of 4-6 nm in the wall interstices (Fig. 9A). It should be noted that gold colloid of 9-8 nm scarcely moved through the wall at any P (Fig. 9B).

How could large dextrans move into a wall that excluded gold of similar dimensions? Part of the mechanism might be
explained by the hydrodynamic nature of the Stokes’ diameter for dextrans, which are derived from relatively dilute polymer solutions (Williams et al., 1998). Axial asymmetry of dextran in solution appears to be significant (Scott et al., 2000), and a small molecular cross-section might allow entry into the matrix while the larger axial dimension might be excluded (Muthukumar, 2003; Wei and Srivastava, 2003). Laurent and Killander (1964) visualized soluble dextrans as flexible rods thickened by branching of the dextran. With the branching, they determined a typical cross-sectional diameter of 0.7 nm for the rods, which is much smaller than the interstices in the wall matrix in the present work. When P concentrated the smaller dextrans against the inner wall face, the polymer ends would be similarly concentrated, and end-first entry would accelerate. By contrast, gold colloids would not present a molecular cross-section smaller than the colloid diameter and would be excluded regardless of the concentration.

The asymmetry decreases with molecular weight (Scott et al., 2000) probably because molecule ends become increasingly hidden within the polymer volume. For larger dextrans to enter the wall, the whole molecule or molecular segment would need to be deformed. When P concentrated the dextran, the molecules pressing against each other might have created momentary asymmetry that deformed the molecules sufficiently to enter. Also, the large P difference across the inner face of the matrix would press molecules against the face, providing a mechanical force (Muthukumar, 2003) that might deform the whole molecule or segment of molecule sufficiently. In these ways, the entropic barrier to entry would be overcome and would require more energy than starting a polymer end because the entire polymer or polymer segment would need to assume entropically less favourable conformations (Muthukumar, 1999). Because P and its accompanying concentration effects would supply the energy for large-scale deformation, a threshold P would appear above which wall entry could begin.

Once entry began above the threshold, the rest of the molecule could follow from the energy supplied by P and its concentrating effects. Above the threshold, the 70-kDa dextran eventually moved through the wall as rapidly as the 38-kDa dextran. The 70-kDa dextran acted as if, once the initial entropic barrier to entry was overcome, subsequent small-scale molecular unfolding occurred with only a small additional energy, i.e. involving only a few bonds at a time. That this scenario occurred at the inner wall face was shown in Fig. 8B where the major barrier to polymer movement was the initial entry into the wall. The appearance of additional free energy barriers to entry as polymers become larger is predicted by computer simulations that indicate larger segments of the molecule must enter the wall before the entire molecule can successfully deform and escape into the wall (Muthukumar, 2001). The appearance of a molecular weight-dependent P threshold in the present data may be interpreted as experimental confirmation of this prediction.

Other effects on the polymers seem much less likely to account for the accelerated entry caused by P. When high P concentrated the polymer solutions, it could have caused increased viscosity or even transitions to a solid or gel-like state, but these changes would hinder entry, not accelerate it.

After fully entering the matrix, the dextran was totally enclosed by the wall interstices, which were at least 5 μm long (the walls were about 5 μm thick). At this length, the interstices were many times the extended length of the largest dextran used in this work. Because the polymer was subjected only to atmospheric pressure in the matrix, each molecule’s net movement would be directed by diffusion by a reptation-type mechanism (Nykypanchuk et al., 2002) resulting from chain flexibility and strong axial asymmetry due to being constrained by the surrounding matrix (Granath, 1958; Munch et al., 1979; Amsden, 2002). During reptation the rate of translocation could be hindered by transient wall/polymer interactions (Davidson and Deen, 1988), which become significant when polymers diffuse through narrow channels that are longer than the extended polymer. Diffusion rates under these conditions have been observed to be dependent on molecular weight (Meller et al., 2001; Loeble et al., 2003; Slonkina and Kolomeisky, 2003).

Therefore, at the P found in living cells, four kinds of molecular movement into the cell walls could be identified. Small molecules such as inorganic salts or inorganic carbon, sugars, or growth regulators are expected to move readily into the wall matrix where they would be unaffected by P (Fig. 10A). Larger molecules were hindered as their Stokes’ diameters approached the diameter of wall interstices. P increased the rate of polymer entry (Fig. 10B). Polymers with Stokes’ diameters larger than the wall interstices entered by starting a molecular end or fold, then unfolding into the opening aided by concentrating and deforming energy of P (Fig. 10C). Polymers with Stokes’ diameters more than twice the diameter of the wall openings seemed to have their ends obscured in the polymer coils and branches, and P had to deform the whole polymer or large parts of the polymer before wall entry began, causing a threshold P-dependence (Fig. 10D). Non-deformable particles such as gold colloids of the same diameter did not enter the wall (Fig. 10E), indicating that large polymer entry was not caused primarily by enlargement of wall interstices but rather by effects of P on the dimensions of the entering polymers themselves.

The deformation of the dextrans could not have been an artefact from small molecules contaminating the dextrans (Preston et al., 1987) because unconjugated FITC and 4-kDa FITC–dextran moved through the wall relatively easily compared with the larger molecules. If small contaminants had been present in the large FITC–dextrans, movement of the small molecules would have been detected at low P. Further purification or size-fractionation of the dextrans was unnecessary, although polydispersity was probably present involving larger molecular weights than the quoted values. Their presence would not have altered the conclusions.

**Wall deposition and cell growth**

Figure 11 shows a correspondence between the movement of 70-kDa FITC–dextran and the growth rate of a living cell when P was varied in Chara. While the strict
correspondence could be coincidental, it indicates that large polymers would enter the wall more rapidly as \( P \) increases the growth rate, potentially contributing to cell wall enlargement. Typically, the degree of polymerization is large for polysaccharides being delivered to the cell wall. In polymers isolated from pea cell walls, Talbot and Ray (1992) measured molecular weights of 30-kDa for xyloglucans and as high as 1000-kDa for pectins, although the latter tended to aggregate after isolation. Because these matrix polysaccharides are vesicle-derived and synthesized in the protoplasm, they appear to be delivered nearly fully formed to the inner face of the cell wall. In multicellular plants, the degree of polymerization could change because of transglycosidase activity in the walls, especially in the xyloglucan portion of the hemicellulosic matrix (Fry et al., 1992; Nishitani and Tominaga, 1992). However, much of this activity depends on enzyme recognition of specific xyloglucan oligosaccharides that are lacking in \( Chara \) walls (Popper and Fry, 2003). As a result, it seems likely that the molecular weights of FITC–dextran used in the present study would simulate the size and motion of fully formed native matrix polymers delivered to the wall.

CONCLUSIONS

These results establish a role for \( P \) in the deposition of wall polymers in addition to its generally accepted role in extending the walls during the growth process. The fact that similar \( P \) are involved in polysaccharide entry and wall enlargement has important implications for coordinating in vivo wall deposition and growth. \( P \) provides the force for polymer deposition in the wall mainly via a mechanism of concentration and \( P \)-induced deformation of the polymers themselves. The strain of the matrix caused by \( P \), while readily detected and central to wall enlargement, did not account for polymer entry. Polymer deformation allowed entry of polymers larger than the interstices in the surrounding wall. It appears that this \( P \) effect would allow the protoplast to supply large polymers and have them enter the wall or move through where they could play diverse roles.

The selectivity for polymer entry occurred at the inner wall face because steep gradients in concentration and \( P \) existed there and provided the energy for polymer deformation. The threshold \( P \) appeared to increase as molecular weights increased, until normal \( P \) was insufficient to move the polymers into the wall. In this situation, the largest polymers remained pressed against the inner wall face where, in the living cell, they would encase new cellulose microfibrils being produced at the plasma membrane.

ACKNOWLEDGEMENTS

We thank Dr Kirk Czymmek of the Delaware Biotechnology Institute for transmission electron micrographs giving the
dimensions of the gold colloids used in this work, and Professor Norman Wagner of the University of Delaware Department of Chemical Engineering for many stimulating discussions. Early phases of this work were supported by DOE grant DE-FG02-87ER13776 to J.S.B. Later phases were supported by an Okie Fellowship from the College of Marine Studies to T.E.P.

LITERATURE CITED


