

## New g-C<sub>3</sub>N<sub>4</sub>/GO/MoS<sub>2</sub> composites as efficient photocatalyst for photocathodic protection of 304 stainless steel

Hongai Zheng<sup>id</sup><sup>a</sup>, Xin Sun<sup>a</sup>, Yue Liu<sup>b</sup>, Shuangyan Jiang<sup>a</sup>, Derui Wang<sup>a</sup>, Yankun Fan<sup>a</sup>, Lili Hu<sup>a</sup>, Daquan Zhang<sup>a</sup>, Weifeng Yao<sup>a</sup> and Lizhi Zhang<sup>c,\*</sup>

<sup>a</sup> College of Environmental and Chemical Engineering, Shanghai University of Electric Power, Shanghai 200090, China

<sup>b</sup> Department of Production Safety, Shanghai Waterworks Fengxian Co., LTD, Shanghai 201499, China

<sup>c</sup> Department of Orthopedic Surgery, Shanghai YangPu District Central Hospital, YangPu Hospital Affiliated to Tongji University, No. 450 TengYue Road, Shanghai 200090, China

\*Corresponding author. E-mail: drzhanglizhi001@163.com

 HZ, 0000-0002-7162-852X

### ABSTRACT

Photocathodic protection is an economical and environmental metal anticorrosion method. In this research, we successfully synthesized the g-C<sub>3</sub>N<sub>4</sub>/GO (15 wt%)/MoS<sub>2</sub> catalytic materials by a facile hydrothermal method. The results show that the as-prepared g-C<sub>3</sub>N<sub>4</sub>/GO (15 wt%)/MoS<sub>2</sub> composites prominently enhanced photocatalytic activities for the photocathodic protection of 304 stainless steel (SS) compared with the corresponding pristine g-C<sub>3</sub>N<sub>4</sub> and MoS<sub>2</sub>. Notably, the AC impedance results demonstrated that the R<sub>ct</sub> value of 304 SS coupled with g-C<sub>3</sub>N<sub>4</sub>/GO (15 wt%)/MoS<sub>2</sub> decreased to 35.66 Ω•cm<sup>2</sup>, which is 29 and 37 times lower than that of g-C<sub>3</sub>N<sub>4</sub> and MoS<sub>2</sub> alone. In addition, g-C<sub>3</sub>N<sub>4</sub>/GO (15 wt%)/MoS<sub>2</sub> provided the highest current density (77.19 μA•cm<sup>2</sup>) for the 304 SS, which is four times that of pristine g-C<sub>3</sub>N<sub>4</sub>. All results indicate that as-prepared g-C<sub>3</sub>N<sub>4</sub>/GO (15 wt%)/MoS<sub>2</sub> photocatalysts have produced a distinct enhancement on photocathodic protection performance. An optimum decorating amount of MoS<sub>2</sub> onto g-C<sub>3</sub>N<sub>4</sub> forms heterojunctions of g-C<sub>3</sub>N<sub>4</sub>/MoS<sub>2</sub>, which favor the separation of electrons and holes efficiently. Furthermore, the addition of GO further promotes the separation and transfer of photo-induced carriers.

**Key words:** graphitic carbon nitride, heterojunction, photocathodic protection, photoelectrochemical

### HIGHLIGHTS

- g-C<sub>3</sub>N<sub>4</sub>/GO/MoS<sub>2</sub> photocatalytic activity was enhanced to protect 304 stainless steel.
- The absorption boundary of g-C<sub>3</sub>N<sub>4</sub>/GO/MoS<sub>2</sub> was red-shifted toward the visible area.
- The photo-induced carriers' recombination of g-C<sub>3</sub>N<sub>4</sub>/GO/MoS<sub>2</sub> was inhibited.
- The 304 stainless steel was effectively protected when coupled to a photo-anode.

## 1. INTRODUCTION

With the rapid development of industry, energy shortage and environmental pollution have become the vital problems faced by human beings (Gao *et al.* 2018; He *et al.* 2018; Zhou *et al.* 2019). The utilization of solar energy has become a good solution to these serious problems. Therefore, the semiconductor photocatalyst has attracted increasing attention, and includes CO<sub>2</sub> photoreduction (Hashemizadeh *et al.* 2018; Xu *et al.* 2018), hydrogen generation (Mo *et al.* 2019), pollutant degradation (Xie *et al.* 2018), and photocathodic protection (PCP) (Guan *et al.* 2018b). Stainless steel materials can withstand high temperatures, oxidation resistance, and have superior mechanical properties, so they are widely used in industry, especially in environmental protection equipment, including delivery pipes, valves, water treatment equipment, and circulating cooling water systems (Ryan *et al.* 2002). However, stainless steel still has the risk of corrosion. Many factors induce corrosion in the water treatment system, including the microbial community in the sewage, low pH, and high salt concentration (Cui *et al.* 2016). Once the stainless steel device fails, it will cause huge economic losses. Therefore, it is very important to propose a reasonable anti-corrosion method (Liu *et al.* 2014; Chen *et al.* 2019). Compared to many metal anticorrosion methods, such as corrosion inhibitors (Fernandes *et al.* 2019; Sadeghi Erami *et al.* 2019), coating (Kiosidou *et al.* 2018; Rodriguez *et al.* 2018), and cathodic protection of sacrificial anodes (Zhang *et al.* 2012; Guan *et al.* 2018b), PCP is an economical and

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environmental method with promising application prospects. Normally, a semiconductor is coated on the protected metal surface or acts as an anode through wires connected to the protected metal (Ren *et al.* 2016). Electrons of the semiconductor in the valence band (VB) excited by light, migrate into the conduction band (CB), and the corresponding holes generate in the VB (Jing *et al.* 2019). At the interface between semiconductor thin film and the solution, the holes ( $h^+$ ) migrate to the surface of semiconductor particles and oxidize the electron donor (such as  $H_2O$ ,  $OH^-$ ), while electrons ( $e^-$ ) migrate to the protected metal. As a result, the electron density on the surface of the protected metal increases and the corrosion potential shifts negatively so that the metal entered the thermodynamic and thermal stable region to achieve the purpose of cathodic protection. Therefore, in PCP, sufficiently negative CB potential is the postulate (Sun *et al.* 2015).  $TiO_2$  coatings are the most widely used photocatalyst in PCP. But the band gap of  $TiO_2$  is 3.2 eV, which limits its application in visible light range (Guan *et al.* 2018a). Many semiconductor materials have been developed used as photocatalysts in PCP including metal oxides ( $ZnO$  (Liu *et al.* 2019b),  $WO_3$  (Ma *et al.* 2019),  $Fe_2O_3$  (Liu *et al.* 2019a), *et al.*), metal sulfides ( $CdS$  (Vamvasakis *et al.* 2018),  $MoS_2$  (Kong *et al.* 2018), *et al.*) and polymeric ( $g-C_3N_4$  (Xu *et al.* 2019)). In recent years,  $g-C_3N_4$ , as a metal-free photocatalyst, has sparked extensive attention in photocatalysis due to its versatile properties, such as low cost, appropriate band gap (2.7 eV) for visible light absorption, good thermal stability, and optical properties (Wang *et al.* 2008; Ong *et al.* 2016; Wang *et al.* 2018b). However, the high recombination rate of photo-induced electrons and hole pairs is still a barrier for photocatalytic efficiency. A feasible way is constructing a heterojunction to overcome these disadvantages (Jiao *et al.* 2019). Heterojunction refers to the recombination of two semiconductors with matching band structures, one as an acceptor for electrons and the other as an acceptor for holes. When illuminated, the photoelectrons in the CB of one semiconductor will be transferred to the CB of another semiconductor to improve the efficiency of carrier transfer and separation (Li *et al.* 2020). Constructing  $g-C_3N_4$  and other semiconductors (such as  $CdS$ ,  $FeOCl$ ,  $WO_3$  and  $Ag_3PO_4$ ) into a heterojunction framework can not only retain the advantages of each component separately, expand the photoresponse range, but also reduce the recombination of photogenerated electrons and holes, thereby improving photocatalytic performance (Katsumata *et al.* 2013; Du *et al.* 2021; Nguyen *et al.* 2021; You *et al.* 2021). It has been reported that the formation of  $rGO/Fe_2O_3/g-C_3N_4$  nanocomposites exhibited superior photocatalytic ability by enhancing charge migration and separation (Shanavas *et al.* 2019).

Loading noble metals, such as Au and Pt, effectively improve the activity of photocatalysts. Recent studies have shown that  $MoS_2$  can be applied as a substitution for noble metals in the integration of co-catalysts. Sun *et al.* (Liu *et al.* 2018) fabricated  $MoS_2$  quantum dots on  $g-C_3N_4$  and found that the composites markedly improved  $H_2$  evolution activities. The enhancement of photocatalytic  $H_2$  evolution was attributed to the formation of the heterojunction interface. The layered structure of  $MoS_2$  is similar to graphene and can match with that of  $g-C_3N_4$ . Furthermore, researchers have confirmed the heterojunction formation between  $g-C_3N_4$  and  $MoS_2$  (Zhang *et al.* 2019b). Therefore, the  $g-C_3N_4/MoS_2$  heterojunction promotes the migration and separation of photo-induced charge. GO has been introduced as an excellent electron conductive material that could promote the transport of photo-induced electrons and hole pairs in most ternary heterojunctions. Yet, as far as we know, there has been no previous report on  $g-C_3N_4/GO/MoS_2$  as a ternary photocatalyst in PCP.

Based on the above issues, we constructed  $g-C_3N_4/GO/MoS_2$  ternary semiconductor heterojunctions through a facile hydrothermal method, trying to develop a green and economical photocatalyst with good photocatalysis performance. The 304 SS was used to investigate the PCP performance of  $g-C_3N_4/GO/MoS_2$ . The ternary heterojunction composites exhibited high PCP performance. A possible mechanism of the catalyst promoting PCP was proposed.

## 2. EXPERIMENTS

### 2.1. Materials

The model of fluorine-doped tin oxide (FTO) conductive glass used in this experiment was FTO-P003, which was purchased from Zhuhai Kaivo Optoelectronic Technology Co., Ltd. The composition of 304 SS (wt%) is as follows: Fe (71.87), C (0.07), Cr (17.0), Ni (8.0), Mn (2.0), Si (1.0), P (0.035), and S (0.03). All the chemical reagents used in the experiments were analytically pure and were not further purified in the process of use.

### 2.2. Synthesis of layer $g-C_3N_4$

Here, 5 g of melamine was calcinated (heating rate: 2 °C/min) at 550 °C for 3 h in a tube furnace in nitrogen. Afterwards, the resulting bulk  $g-C_3N_4$  was ground into powder in a mortar. Then, 0.5 g of  $g-C_3N_4$  powder was calcinated (heating rate: 5 °C/

min) at 500 °C for 3 h in a tube furnace in air. Finally, the corresponding white layer of g-C<sub>3</sub>N<sub>4</sub> product was collected (Papailias *et al.* 2018).

### 2.3. Synthesis of GO

Graphene oxide (denoted as GO) was prepared through oxidation of the natural graphite powder by a modified Hummers' method (Marinoiu *et al.* 2017, 2020). In detail, the mixture of 3.0 g graphite powder, 3.0 g P<sub>2</sub>O<sub>5</sub>, and 3.0 g K<sub>2</sub>S<sub>2</sub>O<sub>8</sub> was added into 50 mL of sulphuric acid (18.4 M). Next, the solution was heated to 80 °C and kept stirring for 5 h. The mixture was slowly diluted with 300 mL deionized (DI) water. Then the pre-oxidized graphite powder was obtained by filtration. In a next step, the pre-oxidized graphite powder was put into 50 mL of sulphuric acid (18.4 M); 15 g KMnO<sub>4</sub> powder was slowly put under ice batches during stirring. After that, they were heated to 35 °C and kept for 2 h under stirring. After cooling down to room temperature, 20 mL of 30% H<sub>2</sub>O<sub>2</sub> was added dropwise. Following that, the color of the mixture turned to golden yellow. Finally, the mixture was washed with diluted HCl solution (v:v = 1:10) by centrifugation. The as-obtained brown product was washed with DI water until the pH value was about 7.0.

### 2.4. Synthesis of g-C<sub>3</sub>N<sub>4</sub>/GO/MoS<sub>2</sub>

The g-C<sub>3</sub>N<sub>4</sub>/GO/MoS<sub>2</sub> composites were achieved via a hydrothermal method. The amount of GO added was 15 wt%, and the mass ratio of g-C<sub>3</sub>N<sub>4</sub> to MoS<sub>2</sub> was 1: 1. In a typical synthesis procedure, 0.1613 g of Na<sub>2</sub>MoO<sub>4</sub>·2H<sub>2</sub>O and 0.2504 g of C<sub>2</sub>H<sub>5</sub>NS were dissolved into 60 mL DI water under magnetic stirring for 1 h. Subsequently, 2.13 g GO aqueous sol with 0.05 g g-C<sub>3</sub>N<sub>4</sub> was dissolved into 20 mL DI water. The above two types of mixture solution were interfused completely. After that, the corresponding mixture solution was transferred into a 100 mL Teflon-lined autoclave and maintained at 180 °C for 24 h. Finally, the resulting dark precipitate denoting g-C<sub>3</sub>N<sub>4</sub>/GO (15 wt%)/MoS<sub>2</sub> was collected and fully washed with DI water and ethanol (Hu *et al.* 2018). Figure S1 in Supplementary Information is the illustration of the synthetic process for g-C<sub>3</sub>N<sub>4</sub>/GO (15 wt%)/MoS<sub>2</sub>.

### 2.5. Characterization

The crystal structure of g-C<sub>3</sub>N<sub>4</sub>, MoS<sub>2</sub>, and g-C<sub>3</sub>N<sub>4</sub>/GO (15 wt%)/MoS<sub>2</sub> was detected using a Bruker-D8 X-ray diffractometer (XRD, Bruker, Germany) with CuK $\alpha$  radiation. X-ray photoelectron spectroscopy (XPS) was performed using a photoelectron spectrometer (Kalpha, ThermoFisher, USA). Characterization of the morphology of g-C<sub>3</sub>N<sub>4</sub>, MoS<sub>2</sub>, and g-C<sub>3</sub>N<sub>4</sub>/GO (15 wt%)/MoS<sub>2</sub> was performed using a supra 55 field emission scanning electron microscope (FE-SEM, Zeiss, Germany). Transmission electronic microscopy (TEM) images and element mapping of materials were analyzed using a JMT-2100F electron microscope (JEOL, Japan), equipped with a Super-X, energy-dispersive spectrometer (EDS, Bruker Germany). A UV-2550 spectrophotometer (SHIMADZU, Japan) was used to conduct ultraviolet-visible diffuse reflectance absorption spectra (UV-vis) of the samples. The photoluminescence properties of samples were determined using a RF-5301PC spectrofluorophotometer (Shimadzu, Japan) with an excitation wavelength of 400 nm.

### 2.6. Photoelectrochemical experiment

The electrochemical experiment was carried out with a set of self-made electrochemical testing equipments, the schematic diagram of which is shown in Figure S2.

Two beakers acted as photochemical cell and electrochemical cell. The electrolyte in the photochemical cell was 0.1 M Na<sub>2</sub>S and 0.2 M NaOH, and the electrolyte in the electrochemical cell was 3.5 wt% NaCl. The two beakers were connected by a salt bridge.

In the electrochemical cell, the Pt sheet was used as the counter electrode, the calomel electrode was used as the reference electrode, and the 304 SS electrode coupled with the photoelectrode acted as the working electrode. The light source was a 300 W xenon lamp (PLS-SXE300) with a light intensity of 520 MW/cm<sup>2</sup>, which shone directly on the back of the light electrode. The distance between the light source and the light electrode was 5 cm (Zhang *et al.* 2014).

Preparation of the salt bridge: 3 g agar powder was dissolved in 100 mL DI water and heated to 90 °C until completely dissolved, followed by the addition of 34.2 g potassium chloride. Subsequently, the above solution was transferred to a U-shaped tube with a rubber dropper and cooled to room temperature.

Preparation of the 304 SS electrode: the 304 SS was cut into small pieces of 10 mm × 10 mm × 3 mm and polished step by step with sandpaper. A copper wire was welded to the back of the 304 SS sheet, followed by sealing with epoxy resin, and then polished evenly with 2,000 mesh SiC papers, so that the exposed surface (working surface) area of the 304 SS sheet

was 10 mm × 10 mm. Finally, the above electrode was washed in ethanol with ultrasound for 30 min, then rinsed with DI water, and dried in a drying oven (room temperature) for 24 h for later use.

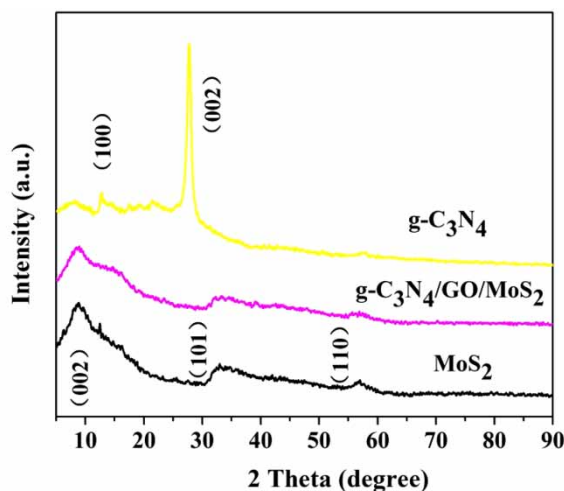
Preparation of the photoelectrode: the FTO was cut into small pieces of 10 mm × 20 mm and put into the cleaning solution (10 mL DI water + 10 mL isopropyl alcohol + 10 mL acetone) with ultrasonic vibration for 30 min, then rinsed with DI water, followed by sealing the resulting FTO surface with tape, leaving only a 10 mm × 10 mm area as a working surface. Afterwards, 20 mg of photocatalyst, 20 μL naphthol, and 1 mL of ethanol were mixed in a centrifuge tube. After ultrasonic (100 W/40 kHz) bombardment for 30 min, the corresponding slurry was collected. Finally, 20 μL of the slurry was smeared onto the working surface of the FTO as evenly as possible to ensure the same sample mass on the FTO surface (about 0.4 mg/cm<sup>2</sup>). After natural air drying, the obtained photoelectrode was heated to 100 °C for 2 h.

CHI660D electrochemical workstation was used to measure the electrochemistry performance of 304 SS coupled with different photocatalysts. Electrochemical impedance spectroscopy (EIS) tests were carried out in the frequency between 10<sup>5</sup> and 10<sup>-2</sup> Hz, and the AC voltage amplitude was 5 mV. Dynamic potential polarization curves (Tafel plot) were tested with a scan rate of 1 mV·s<sup>-1</sup>. The linear sweep voltammetry (*i*-V) measurement was obtained from -1.0 V to 0.4 V (vs. SCE) with a scan rate of 0.02 mV·s<sup>-1</sup>. The time gap between light and dark was 10 s. The CV curves test was performed with a scan rate of 10 mV. The Mott-Schottky curves were tested from -1.0 to 0.0 V (vs. SCE). The AC voltage amplitude was 10 mV, and the test frequency was 1,000 Hz.

### 3. RESULTS AND DISCUSSION

#### 3.1. Structure and morphology analysis

The crystal structure of g-C<sub>3</sub>N<sub>4</sub>, MoS<sub>2</sub>, and g-C<sub>3</sub>N<sub>4</sub>/GO (15 wt%)/MoS<sub>2</sub> was analyzed by XRD. From Figure 1, we can see that at 13.0° and 27.4°, the characteristic diffraction peaks of g-C<sub>3</sub>N<sub>4</sub> exhibited corresponding to (100) and (002) planes (JCPDS 87-1526) (Yang *et al.* 2020). The weak peak at 13.0° was derived from the in-plane structural repeated motif of tri-s-triazine units with an interplanar distance of  $d = 0.663$  nm (Wang *et al.* 2018c). The dominant peak, which was at 27.4°, was derived from the aromatic ring system stack whose interplanar distance was 0.324 nm (Wang *et al.* 2018a). As for MoS<sub>2</sub> samples, the two diffraction peaks at 33.3° and 58.8° corresponded well to MoS<sub>2</sub> (JCPDS 37-1492) (Zhang *et al.* 2019a), which represented the (101) and (110) crystal planes, respectively. The 13.9° diffraction peak corresponding to the (002) crystal plane appeared at 10.3°, which was due to its interlayer expansion effect (Zhang *et al.* 2020). According to the Bragg equation:  $2dsin\theta = \lambda$ , the interlayer spacing of the (002) plane of the MoS<sub>2</sub> sample could be calculated. The interplanar distance was about 0.86 nm. In g-C<sub>3</sub>N<sub>4</sub>/GO (15 wt%)/MoS<sub>2</sub> composites, the main performance was the characteristic peak of MoS<sub>2</sub>, and the peak height was decreased. This was because the sample contains g-C<sub>3</sub>N<sub>4</sub> and GO, which resulted in a decrease in its crystallinity. The reason for the weak characteristic peak of g-C<sub>3</sub>N<sub>4</sub> was that the high crystalline MoS<sub>2</sub> grew on the surface of g-C<sub>3</sub>N<sub>4</sub> (Fageria *et al.*



**Figure 1** | XRD patterns of g-C<sub>3</sub>N<sub>4</sub>, MoS<sub>2</sub>, and g-C<sub>3</sub>N<sub>4</sub>/GO (15 wt%)/MoS<sub>2</sub>.



2017). In addition, no characteristic peak of GO was observed in g-C<sub>3</sub>N<sub>4</sub>/GO (15 wt%)/MoS<sub>2</sub> samples, which was due to the relatively low GO content (Guan *et al.* 2017).

XPS spectra recorded the surface electronic states and chemical composition of ternary composites. Figure S3a indicated the spectrum of g-C<sub>3</sub>N<sub>4</sub>/GO (15 wt%)/MoS<sub>2</sub> demonstrated the co-existence of Mo, S, C, O, and N elements. As shown in Figure S3b, the peak at 284.16 eV showed the occurrence of sp<sup>2</sup> C = C bonds of g-C<sub>3</sub>N<sub>4</sub> (Shao *et al.* 2016). The peak at 285.64 eV can be consistent with sp<sup>3</sup> hybridized C atoms (C-(N)<sub>3</sub>) (Fu *et al.* 2018). The peak located at 288.20 eV was ascribed to the sp<sup>2</sup> carbon N-C = N, which was in the N-containing aromatic rings (Fu *et al.* 2018). Figure S3c shows N 1s spectrum, the peak at 398.49 eV was attributed to C-N = C groups. The peak at 400.23 eV showed the occurrence of the tertiary nitrogen N-(C)<sub>3</sub> groups (Sun *et al.* 2020). In addition, the peak at 404.46 eV may be associated with positive charge localization or the charging effects in the heterocycles (Wang *et al.* 2020). As can be seen in Figure S3d, the peak centered located at 531.53 eV correspond to O 1s orbital (Hu *et al.* 2018). In Figure S3e, the S 2p peaks, which were located at 161.53 eV and 162.95 eV, were consistent with S 2p<sub>3/2</sub> and S 2p<sub>1/2</sub> of S<sup>2-</sup>, respectively (Guan *et al.* 2017). In addition, the peak located at 164.23 eV corresponds to the C-S bond formed by replacing sulfur with lattice nitrogen in the g-C<sub>3</sub>N<sub>4</sub> framework (Bai *et al.* 2018). Figure S3f shows the Mo 3d high-resolution XPS spectrum of the g-C<sub>3</sub>N<sub>4</sub>/GO (15 wt%)/MoS<sub>2</sub> sample. The peaks located at 232.41 eV and 228.89 eV corresponded to Mo 3d<sub>3/2</sub> and Mo 3d<sub>5/2</sub> of Mo<sup>4+</sup> in MoS<sub>2</sub>, respectively (Zhou *et al.* 2020). The peak at 235.65 eV was consistent with the Mo atoms exposed on the surface (Ou *et al.* 2020). The weak peak at 226.03 eV can be ascribed to S<sup>2-</sup> and Mo<sup>4+</sup>, which is the typical peak for MoS<sub>2</sub> (Bai *et al.* 2018).

The morphology of the ternary composites was further characterized by SEM. The g-C<sub>3</sub>N<sub>4</sub> exhibited an average of 100 nm thicknesses with a lamellar stackable structure. In the MoS<sub>2</sub>/GO composites, MoS<sub>2</sub> was in the form of nanoparticles, and the wrinkle GO was enclosed on the surface of MoS<sub>2</sub> (Figure S4a and 4b). As for the sample, the wrinkles GO wrapped MoS<sub>2</sub> nanoparticles attached to the surface of the g-C<sub>3</sub>N<sub>4</sub> sheet layer (Figure S4c). Figure S4d indicated the co-existence of C, N, O, S, and Mo elements in the g-C<sub>3</sub>N<sub>4</sub>/GO (15 wt%)/MoS<sub>2</sub> sample.

The morphologies of g-C<sub>3</sub>N<sub>4</sub>/GO (15 wt%)/MoS<sub>2</sub> samples were further observed by TEM. MoS<sub>2</sub> was a petal-shaped nanosheet. MoS<sub>2</sub> nanosheets grew on g-C<sub>3</sub>N<sub>4</sub> (Figure S5a). Figure S5b shows the HR-TEM images of g-C<sub>3</sub>N<sub>4</sub>/GO (15 wt %)/MoS<sub>2</sub>. Each petal of MoS<sub>2</sub> nanosheets displayed obvious lattice fringes with a lattice spacing of 0.86 nm, corresponding to the (002) crystal plane of MoS<sub>2</sub>. MoS<sub>2</sub> and g-C<sub>3</sub>N<sub>4</sub> were closely attached, and the crystal lattices of both could be clearly seen. Figure S5c further demonstrates the co-existence of C, N, O, S, and Mo elements in the g-C<sub>3</sub>N<sub>4</sub>/GO (15 wt%)/MoS<sub>2</sub> sample.

### 3.2. Optical properties

The UV-vis DRS of g-C<sub>3</sub>N<sub>4</sub>, MoS<sub>2</sub>, g-C<sub>3</sub>N<sub>4</sub>/MoS<sub>2</sub>, g-C<sub>3</sub>N<sub>4</sub>/GO (15 wt%)/MoS<sub>2</sub> are depicted in Figure S6. The visible light absorption threshold of g-C<sub>3</sub>N<sub>4</sub> was 410 nm. MoS<sub>2</sub> samples were black and therefore absorbed in the visible range (Zhang *et al.* 2019b). After adding MoS<sub>2</sub>, the g-C<sub>3</sub>N<sub>4</sub>/MoS<sub>2</sub> composites showed strong light absorption ability, and the heterostructure formed between them. g-C<sub>3</sub>N<sub>4</sub>/GO (15 wt%)/MoS<sub>2</sub> had stronger absorption in the visible light region than g-C<sub>3</sub>N<sub>4</sub>/MoS<sub>2</sub> composites. This was due to the addition of GO, which absorbed light in the full spectral range, thereby increasing the visible light absorption of the composite. Using the Kubelka–Munk equation, the Eg of the obtained sample was calculated:

$$ahv = A(hv - Eg)^n \quad (1)$$

where  $\alpha$ ,  $h$ ,  $v$ ,  $A$ , and  $Eg$  represent absorption coefficient, Planck constant, light frequency, constant and band-gap energy, respectively. The value of  $n$  depends on the transition characteristic of the material,  $n = 1/2$  for direct transition and  $n = 2$  for indirect transition. g-C<sub>3</sub>N<sub>4</sub> and MoS<sub>2</sub> were indirect transitions photocatalysts. The band gap of g-C<sub>3</sub>N<sub>4</sub> was calculated to be 2.81 eV. For MoS<sub>2</sub>, g-C<sub>3</sub>N<sub>4</sub>/MoS<sub>2</sub> and g-C<sub>3</sub>N<sub>4</sub>/GO (15 wt%)/MoS<sub>2</sub> samples, the band gaps were calculated as 1.51 eV, 1.71 eV, and 1.87 eV, respectively.

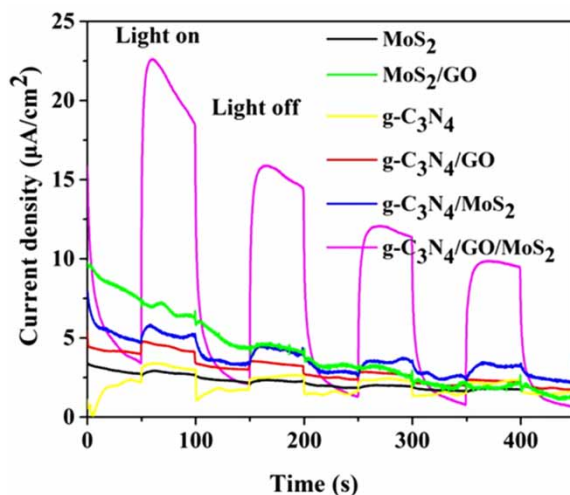
At an excitation wavelength of 400 nm, the PL spectra of MoS<sub>2</sub>, g-C<sub>3</sub>N<sub>4</sub>/MoS<sub>2</sub> and g-C<sub>3</sub>N<sub>4</sub>/GO (15 wt%)/MoS<sub>2</sub> composites are displayed in Figure S7. Two emission peaks were observed, which appeared at 450 and 470 nm (Liang *et al.* 2020). The emission peak of MoS<sub>2</sub> was the strongest, and indicated that the high recombination efficiency of photo-generated carriers. After adding g-C<sub>3</sub>N<sub>4</sub>, the peak value decreased, indicating that the separation of photo-generated carriers had been promoted. The emission peak of g-C<sub>3</sub>N<sub>4</sub>/GO (15 wt%)/MoS<sub>2</sub> composites was the weakest, indicating that the addition of GO promoted

the transfer of photogenerated electrons and holes. Therefore, the  $g\text{-C}_3\text{N}_4/\text{GO}$  (15 wt%)/ $\text{MoS}_2$  ternary composites showed the strongest photoelectrochemical activity.

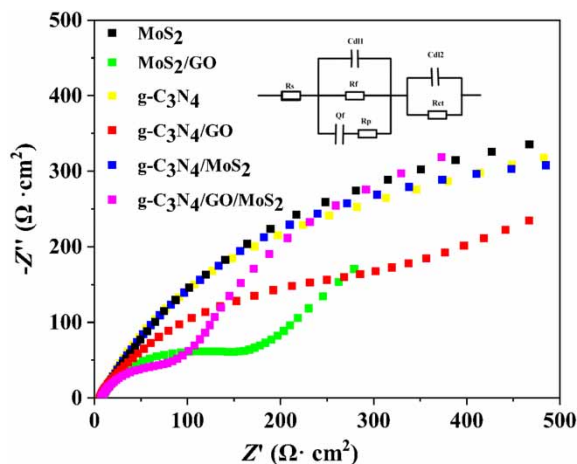
### 3.3. Photocathodic protection performance

To investigate the photoelectrochemical performance of the composite catalyst, photogenerated current density ( $I$ - $T$ ) was performed at interval of 50 s/time illumination. As shown in Figure 2, after illumination, the photocurrent density of the samples increased. The greater the increase in photogenerated current density, the higher was the charge separation efficiency of the catalyst. The photocurrent of  $\text{MoS}_2$  alone was relatively low, and the photocurrent significantly increased after being combined with  $g\text{-C}_3\text{N}_4$ , which reached  $6 \mu\text{A}/\text{cm}^2$ .  $g\text{-C}_3\text{N}_4/\text{GO}$  (15 wt%)/ $\text{MoS}_2$  has the best photocurrent response, which can reach  $22 \mu\text{A}/\text{cm}^2$ . This is because the heterostructure of  $g\text{-C}_3\text{N}_4/\text{MoS}_2$  increased the electron-hole separation efficiency, and the addition of GO was more conducive to the photogenerated electron transfer between  $g\text{-C}_3\text{N}_4/\text{MoS}_2$ .

EIS test is an effective method to analyze and evaluate the PCP performance of samples. Figure 3 shows the EIS Nyquist plots and equivalent circuit diagram of 304 SS coupled with different photoelectrodes in light. The 304 SS coupled with  $g\text{-C}_3\text{N}_4/\text{GO}/\text{MoS}_2$



**Figure 2** | Photogenerated current density of  $g\text{-C}_3\text{N}_4$ ,  $\text{MoS}_2$ , and  $g\text{-C}_3\text{N}_4/\text{GO}$  (15 wt%)/ $\text{MoS}_2$  composites photocatalytic under intermittent visible light.



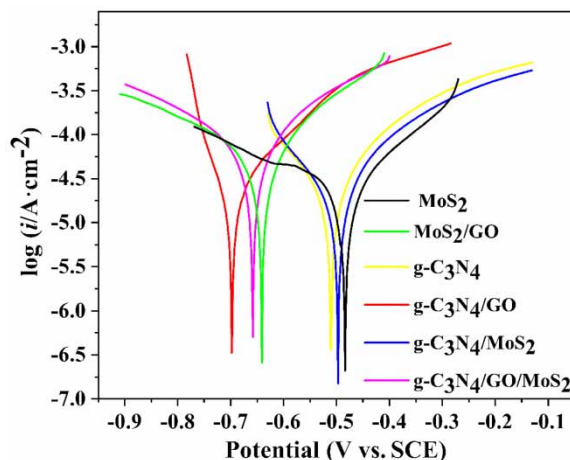
**Figure 3** | (a) Nyquist plots of 304 SS coupled with the photoelectrodes under illumination conditions. The inset is the equivalent circuit by EIS results fitting.

$\text{C}_3\text{N}_4/\text{MoS}_2$  showed an obvious and smaller capacitive reactance arc than pure  $\text{g-C}_3\text{N}_4$  and  $\text{MoS}_2$ . After adding GO, two arcs appeared in the low-frequency region. The small semi-circular arc indicated low charge transfer resistance (Wei *et al.* 2019). As shown in Figure 3, the equivalent circuit as R (CR(QR)) (CR) fitting circuit was adopted to fit the EIS results using the ZSimpWin software, where  $C_{dl}$  and  $Q_f$  represent the double-layer capacitance and the capacitance of the surface film, respectively. Furthermore,  $R_s$ ,  $R_f$ ,  $R_p$ , and  $R_{ct}$  represent the electrolyte solution resistance, the resistance of oxide films, electrolyte, and charge transfer, respectively.  $Y$  and  $n$  are constants that are independent of frequency. Detailed fitting data of the impedance spectrum of the material are shown in Table 1.  $R_{ct}$  values can be used to characterize the charge transfer resistance of semiconductor materials. The  $R_{ct}$  coupled with  $\text{g-C}_3\text{N}_4$  and  $\text{MoS}_2$  was  $1,032 \Omega \cdot \text{cm}^2$  and  $1,324 \Omega \cdot \text{cm}^2$ , respectively. The  $R_{ct}$  coupled with  $\text{g-C}_3\text{N}_4/\text{MoS}_2$  was  $967.9 \Omega \cdot \text{cm}^2$ , indicating that the heterostructure reduced the charge transfer resistance of the composites. After adding GO, the  $R_{ct}$  values decreased significantly. The decrease in  $R_{ct}$  value indicated the higher photogenerated electrons transfer. The EIS results showed that the  $\text{g-C}_3\text{N}_4/\text{GO}$  (15 wt%)/ $\text{MoS}_2$  composites transferred a large number of electrons to the surface of 304 SS, resulting in the best photochemical cathodic protection performance.

Figure 4 demonstrates the Tafel plot of 304 SS coupled with kinds of photoelectrodes in light. The corresponding parameters calculated are displayed in Table 2. The potential of 304 SS coupled with  $\text{g-C}_3\text{N}_4/\text{MoS}_2$  was between that of  $\text{g-C}_3\text{N}_4$  and  $\text{MoS}_2$ . After adding GO, the corrosion potentials of 304 SS coupled with all three samples significantly shifted negatively. 304 SS coupled with  $\text{g-C}_3\text{N}_4/\text{GO}$  showed the most negative corrosion potential. As for the corrosion current,  $\text{g-C}_3\text{N}_4/\text{MoS}_2$  was higher than that of  $\text{MoS}_2$  and  $\text{g-C}_3\text{N}_4$  alone, which is ascribed to the formation of heterojunctions. The increase in the corrosion current increased the electron transfer rate (Qiu *et al.* 2020). After the introduction of GO, the corrosion

**Table 1** | Fitting impedance parameters of Nyquist plots of composites

Samples	$R_s$ ( $\Omega \cdot \text{cm}^2$ )	$C_{dl1}$ (F)	$R_f$ ( $\Omega \cdot \text{cm}^2$ )	$Q_f$		$R_p$ ( $\Omega \cdot \text{cm}^2$ )	$C_{dl2}$ (F)	$R_{ct}$ ( $\Omega \cdot \text{cm}^2$ )
				$y$ ( $\mu\text{S} \cdot \text{s}^n \cdot \text{cm}^{-2}$ )	$n$			
$\text{g-C}_3\text{N}_4$	5.495	$8.452 \times 10^{-7}$	691.5	$7.318 \times 10^{-4}$	0.7553	1.411	$9.654 \times 10^{-3}$	1,032
$\text{MoS}_2$	3.932	$2.875 \times 10^{-7}$	848.6	$5.906 \times 10^{-4}$	0.7395	3.916	$9.105 \times 10^{-3}$	1,324
$\text{g-C}_3\text{N}_4/\text{GO}$	6.013	$2.712 \times 10^{-7}$	495.6	$6.188 \times 10^{-4}$	0.634	0.7813	$8.825 \times 10^{-3}$	552.8
$\text{MoS}_2/\text{GO}$	3.807	$1.571 \times 10^{-7}$	190.2	$5.523 \times 10^{-4}$	0.7445	4.272	$1.342 \times 10^{-2}$	359.4
$\text{g-C}_3\text{N}_4/\text{MoS}_2$	4.362	$3.569 \times 10^{-7}$	750.6	$5.907 \times 10^{-4}$	0.7533	2.93	$9.564 \times 10^{-3}$	967.9
$\text{g-C}_3\text{N}_4/\text{GO}$ (15 wt%)/ $\text{MoS}_2$	2.604	$8.916 \times 10^{-8}$	3,690	$2.997 \times 10^{-3}$	0.5715	4.463	$2.408 \times 10^{-4}$	35.66



**Figure 4** | Tafel plots of 304 SS coupled with photoanodes under illumination.

**Table 2** | Electrochemical parameters obtained by the Tafel curves of 304 SS coupled with photoanodes under illumination

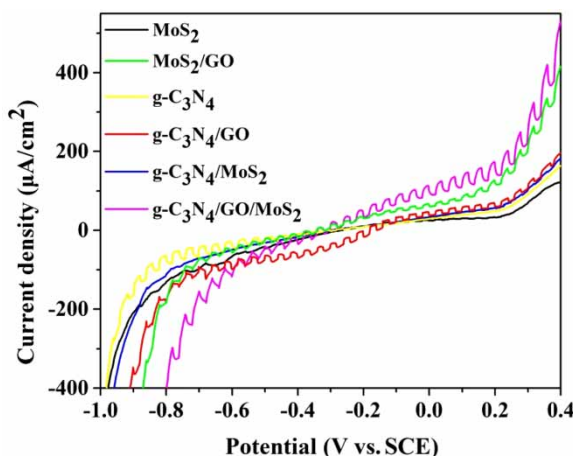
Samples	$E_{\text{corr}}$ (V vs.SCE)	$\beta_c$ (mv dec <sup>-1</sup> )	$\beta_a$ (mv dec <sup>-1</sup> )	$J_{\text{corr}}$ ( $\mu\text{A cm}^{-2}$ )
MoS <sub>2</sub>	-0.484	32.65	54.70	10.74
MoS <sub>2</sub> /GO	-0.641	36.87	60.82	62.97
g-C <sub>3</sub> N <sub>4</sub>	-0.511	75.46	92.35	17.49
g-C <sub>3</sub> N <sub>4</sub> /GO	-0.698	31.69	57.62	35.59
g-C <sub>3</sub> N <sub>4</sub> /MoS <sub>2</sub>	-0.497	76.61	94.83	18.95
g-C <sub>3</sub> N <sub>4</sub> /GO (15 wt%)/MoS <sub>2</sub>	-0.658	40.47	56.08	77.19

currents of 304 SS coupled with g-C<sub>3</sub>N<sub>4</sub> and MoS<sub>2</sub> significantly increased. GO promoted the transfer of photo-generated electrons and holes. 304 SS coupled with g-C<sub>3</sub>N<sub>4</sub>/GO (15 wt%)/MoS<sub>2</sub> revealed the highest current density, which was 77.19  $\mu\text{A}\cdot\text{cm}^{-2}$ . The heterojunction of g-C<sub>3</sub>N<sub>4</sub>/MoS<sub>2</sub> promoted the separation of photogenerated electrons and holes, and the addition of GO promoted the transfer of photo-generated electrons and holes.

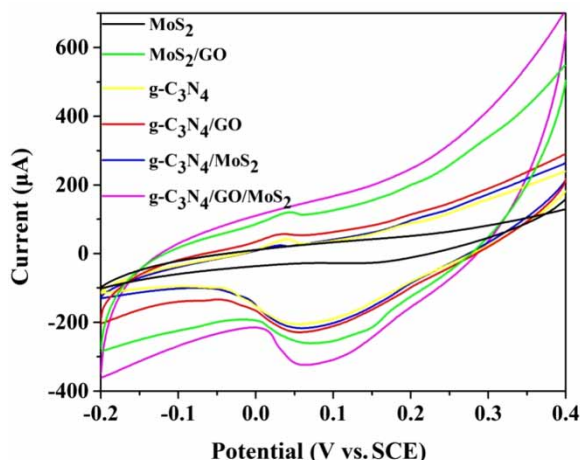
### 3.4. Photoelectrochemical measurements

To study the photoelectrochemical performance of the composite photocatalyst further, the photoinduced *i-V* curves of the composites between -1.0 and 0.4 V were measured under intermittent light. The photoelectrochemical properties of the composites were tested by intermittent illumination of 10 s (Figure 5). The samples generated the zigzag photocurrent response between -1.0 and 0.4 V. Comparing the absolute value of photocurrent of different samples, the photo-generated current density produced by g-C<sub>3</sub>N<sub>4</sub>/GO (15 wt%)/MoS<sub>2</sub> was the highest. The g-C<sub>3</sub>N<sub>4</sub>/GO (15 wt%)/MoS<sub>2</sub> showed the best photoelectrochemical properties.

The CV curves of the 304 SS coupled with samples were also tested under light. The results are shown in Figure 6. In this experiment, the CV curves of 304 SS coupled with different photocatalysts were tested at a scan rate of 10 mv. The potential ranges of the samples were approximately -0.2 V to 0.4 V. 304 SS coupled with MoS<sub>2</sub> showed double capacitance characteristics. The cathode and anode peaks of other photocatalysts were not the same (Fageria *et al.* 2017). According to the Randles-Sevcik equation, the electroactive surface area (ECSA) was proportional to the peak of the cathode and anode (Shi & Zhao 2014). The larger ECSA means more active sites for the electrochemical reaction. The high cathode and anode peak value generated a larger ECSA. In addition, it exhibited better electrochemical performance. In Figure 6, we can see that the 304 SS coupled with g-C<sub>3</sub>N<sub>4</sub>/GO (15 wt%)/MoS<sub>2</sub> showed the highest peak in the cathode and anode curves. Therefore, the ternary photocatalyst has once again been proved to have the best electrochemical performance.

**Figure 5** | Photoinduced I-V curves of g-C<sub>3</sub>N<sub>4</sub>, MoS<sub>2</sub>, and g-C<sub>3</sub>N<sub>4</sub>/GO (15 wt%)/MoS<sub>2</sub> composites under intermittent visible light.





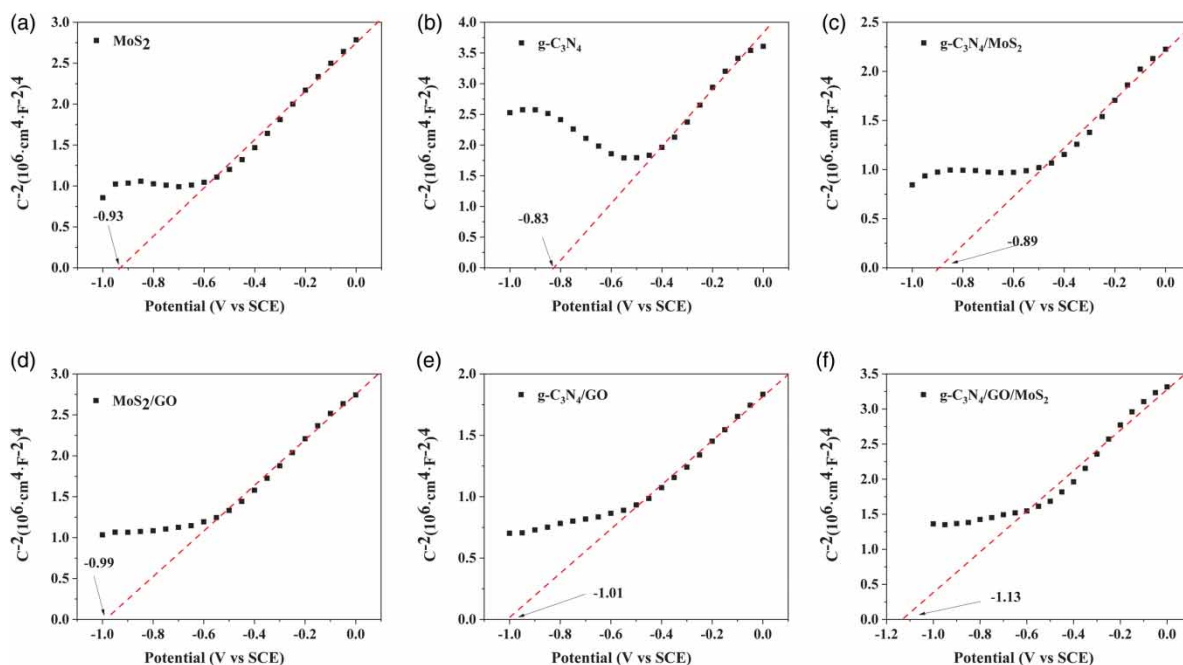
**Figure 6** | CV curves of 304 SS coupled with g-C<sub>3</sub>N<sub>4</sub>, MoS<sub>2</sub>, and g-C<sub>3</sub>N<sub>4</sub>/GO (15 wt%)/MoS<sub>2</sub> composites under illumination.

### 3.5. Mechanism

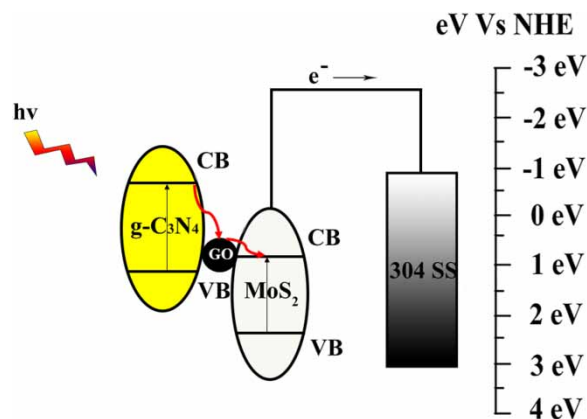
To explore the mechanism of PCP on composites photocatalyst, Mott-Schottky plots were performed. According to the formula listed, we calculated the flat band potential as follows:

$$\frac{1}{C^2} = 2(e\varepsilon\varepsilon_0N_D)^{-1} \cdot \left( E - E_{\text{FB}} - \frac{kT}{e} \right) \quad (2)$$

$C$  represented the capacitance of the space charge region.  $e$  was the electronic charge.  $\varepsilon$  was the relative dielectric constant of the semiconductor.  $\varepsilon_0$  represented the permittivity for free space.  $N_D$  was the charge carrier density.  $E$  was the applied potential.  $E_{\text{FB}}$  was the flat band potential.  $T$  was the temperature.  $k$  was the Boltzmann constant (Jing *et al.* 2019). The flat band potential ( $E_{\text{FB}}$ ) was determined from the horizontal intercept. As shown in Figure 7, the  $E_{\text{FB}}$  of MoS<sub>2</sub> was about  $-0.93$  eV. After adding GO, its  $E_{\text{FB}}$  was negatively shifted to  $-0.99$  eV. The  $E_{\text{FB}}$  of g-C<sub>3</sub>N<sub>4</sub> was about  $-0.83$  eV. After



**Figure 7** | Mott-Schottky plots of g-C<sub>3</sub>N<sub>4</sub>, MoS<sub>2</sub>, and g-C<sub>3</sub>N<sub>4</sub>/GO (15 wt%)/MoS<sub>2</sub> composites under illumination.



**Figure 8** | The photocathodic protection mechanism of g-C<sub>3</sub>N<sub>4</sub>/GO (15 wt%)/MoS<sub>2</sub> composites.

adding GO, its  $E_{FB}$  was negatively shifted to  $-1.01$  eV. The negative shift of  $E_{FB}$  increases the Fermi level of the material, which was more favorable for the transfer of photogenerated electrons (Liu *et al.* 2018; Ou *et al.* 2020). Through GO modification, the  $E_{FB}$  of a photocatalyst was negatively shifted, which improved the transfer efficiency of photogenerated electrons and holes, thereby improving the photocathodic protection ability of the composites. The  $E_{FB}$  of g-C<sub>3</sub>N<sub>4</sub>/MoS<sub>2</sub> was  $-0.89$  eV, which was between the MoS<sub>2</sub> and g-C<sub>3</sub>N<sub>4</sub>. After GO was compounded with g-C<sub>3</sub>N<sub>4</sub>/MoS<sub>2</sub>, the  $E_{FB}$  was negatively shifted to  $-1.13$  eV. This indicated that g-C<sub>3</sub>N<sub>4</sub>/GO (15 wt%)/MoS<sub>2</sub> had better photogenerated charge separation efficiency and more excellent photocathodic protection performance.

Based on the results of the aforementioned experiments, the possible mechanism of the catalyst promoting PCP is proposed in Figure 8. Under visible light, the CB of g-C<sub>3</sub>N<sub>4</sub> and MoS<sub>2</sub> both generated photogenerated electrons and left the photogenerated holes on the VB. In the g-C<sub>3</sub>N<sub>4</sub>/GO (15 wt%)/MoS<sub>2</sub> sample, g-C<sub>3</sub>N<sub>4</sub> and MoS<sub>2</sub> formed the heterojunction. Photogenerated electrons transferred from the CB of g-C<sub>3</sub>N<sub>4</sub> to the CB of MoS<sub>2</sub>. The formation of the heterojunction electric field promoted the separation of photogenerated electrons and holes. The addition of GO further promoted the separation and transfer of photogenerated electrons and holes and simultaneously reduced the Fermi level of the composites. The photoelectrons on the surface of the samples transferred to the 304 SS surface through the wire, which made the self-corrosion potential of 304 SS drop sharply and resulted in the cathodic polarization of 304 SS.

#### 4. CONCLUSION

In this research, the g-C<sub>3</sub>N<sub>4</sub>/GO (15 wt%)/MoS<sub>2</sub> composites were successfully synthesized by a facile hydrothermal method. MoS<sub>2</sub> was a petal-shaped nanosheet characterized by SEM and TEM. Wrinkled GO wrapped MoS<sub>2</sub> nanoparticles attached to the surface of the g-C<sub>3</sub>N<sub>4</sub> sheet. The EIS results depicted that the  $R_{ct}$  value of 304 SS coupled with g-C<sub>3</sub>N<sub>4</sub>/MoS<sub>2</sub> decreased from 1,032 to 967.9  $\Omega \cdot \text{cm}^2$  compared with g-C<sub>3</sub>N<sub>4</sub> alone. Notably, the  $R_{ct}$  value of 304 SS coupled with g-C<sub>3</sub>N<sub>4</sub>/GO (15 wt%)/MoS<sub>2</sub> composites decreased to 35.66  $\Omega \cdot \text{cm}^2$ , which was 29 and 37 times lower than that of g-C<sub>3</sub>N<sub>4</sub> and MoS<sub>2</sub> alone.

The polarization curve fitting values exhibited that g-C<sub>3</sub>N<sub>4</sub>/GO (15 wt%)/MoS<sub>2</sub> provided the highest current density (77.19  $\mu\text{A} \cdot \text{cm}^2$ ) for the 304 SS, which was four times higher than that of single g-C<sub>3</sub>N<sub>4</sub>. The mechanism is that, under light conditions, photo-generated electrons were generated from the CB of g-C<sub>3</sub>N<sub>4</sub> and transferred to the VB of MoS<sub>2</sub>, and the heterojunction formed between g-C<sub>3</sub>N<sub>4</sub> and MoS<sub>2</sub>. The formation of a heterojunction electric field promoted the separation of photogenerated electrons and holes. The addition of GO further promoted the separation and transfer of photogenerated electrons and holes and simultaneously reduced the Fermi level of the composites. Therefore, the ternary composite improved the photocathode protection performance for 304 SS. This research is expected to provide new insights to explore high-efficiency photocatalysts and new corrosion protection technologies with potential application prospects.

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## DECLARATION OF INTERESTS

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

## DATA AVAILABILITY STATEMENT

All relevant data are included in the paper or its Supplementary Information.

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